

# Renormalization of Atomic-Scale Binding-Energy Relations

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# Stress-Corrosion Cracking

- Engineering model:
  - *Cohesive models of fracture*
  - *Coupling of cohesive models and surface chemistry*
  - *Adaptive FE stress analysis*
  - *FE analysis of impurity diffusion:*
    - *Stress-assisted diffusion*
    - *Opening-dependent mixed boundary conditions over cohesive zone*
- Goal: To predict **time to failure** of components under general load histories, environments.

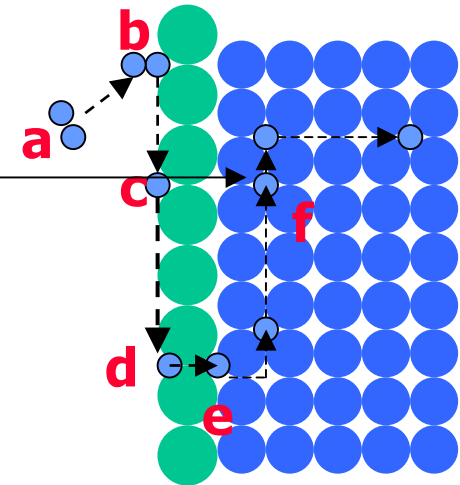


# Hydrogen embrittlement

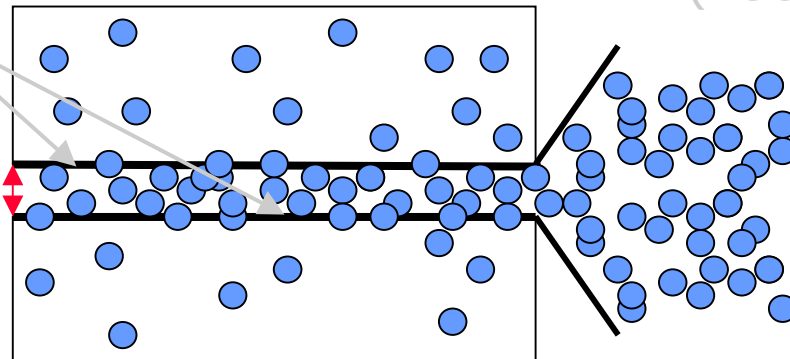
- Hydrogen adsoption
- Hydrogen diffusion
- Free energy density =  $f(T, \Gamma^\pm, \delta)$
- $df(\delta, \Gamma^\pm) = \sigma d\delta + \mu d\Gamma$

$\Gamma^\pm$  = Surface concentrations

$\delta$  = Opening displacement



(Nelson 1972)

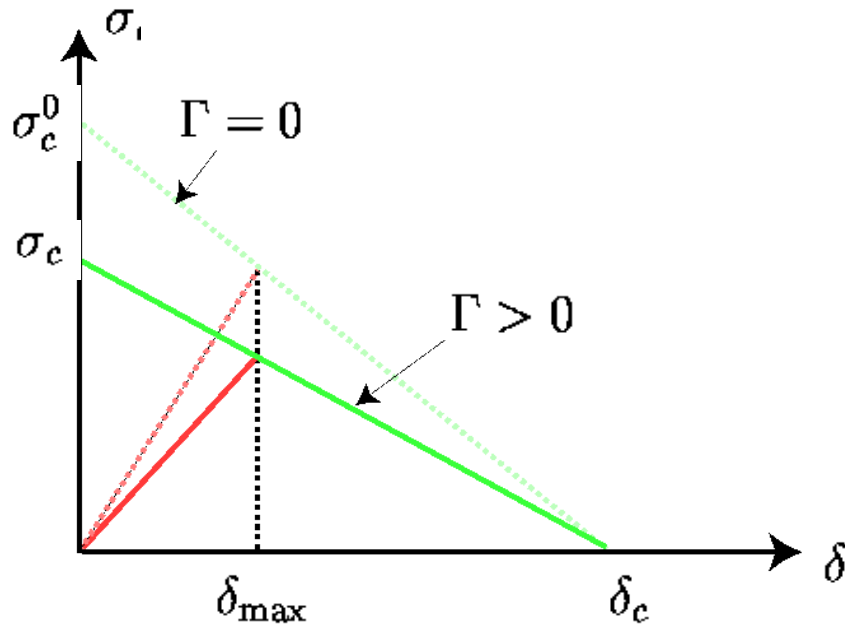


Environment  
defined by  $\mu_{\text{ext}}$

Schematic of interface/impurity system  
(Rice & Wang, 1989)



# SCC – Cohesive model



- Segregant embrittlement:  
(Wang and Rice, 1989):

$$2\gamma_s(\Gamma) = 2\gamma_s^0 - (\Delta g_b^0 - \Delta g_s^0) \Gamma$$

- Effect on cohesive law:

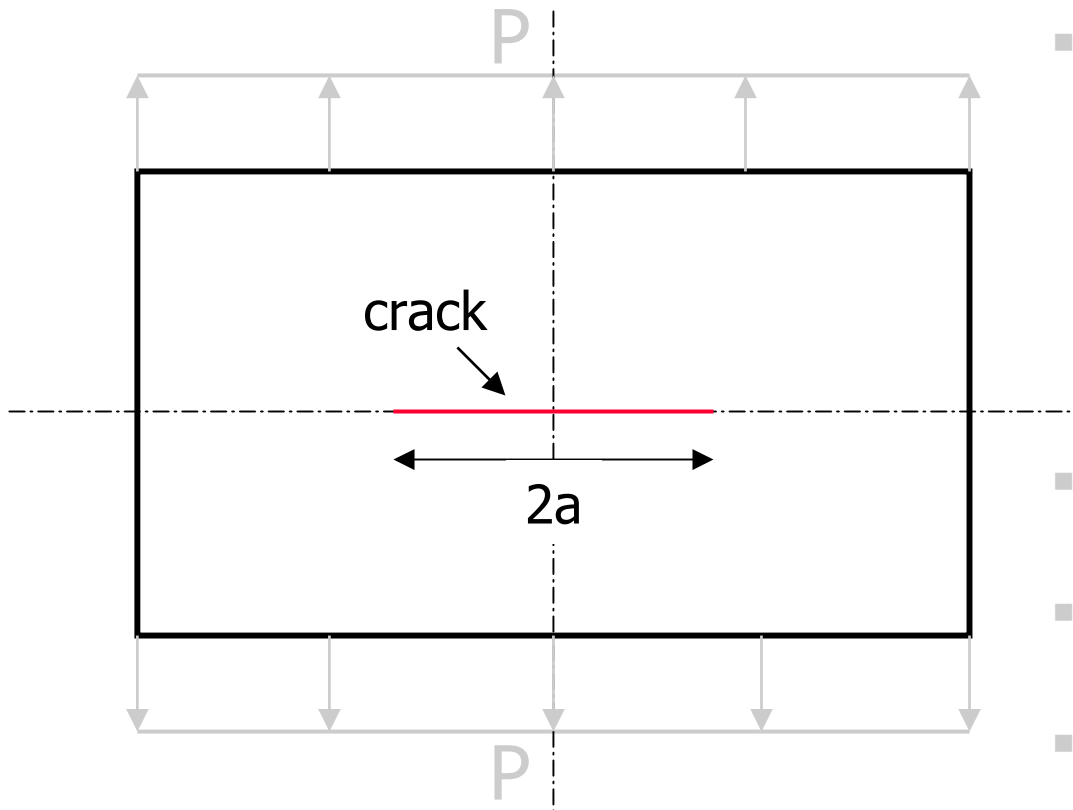
$$\sigma_c(\Gamma) = \sigma_c^0 - \frac{(\Delta g_b^0 - \Delta g_s^0)}{\delta_c} (\Gamma^+ + \Gamma^-)$$

Impurity	$-\Delta g_s^0$	$-\Delta g_b^0$	$\Delta g_b^0 - \Delta g_s^0$
C	73 to 85	50 to 75	-2 to 35
P	76 to 80	32 to 41	35 to 48
Sb	83 to 130	8 to 25	58 to 122
S	165 to 190	50 to 58	107 to 140

(Steel, 300K, g in KJ/mol; after Wang and Rice, 1989)



# SCC – Experimental Validation

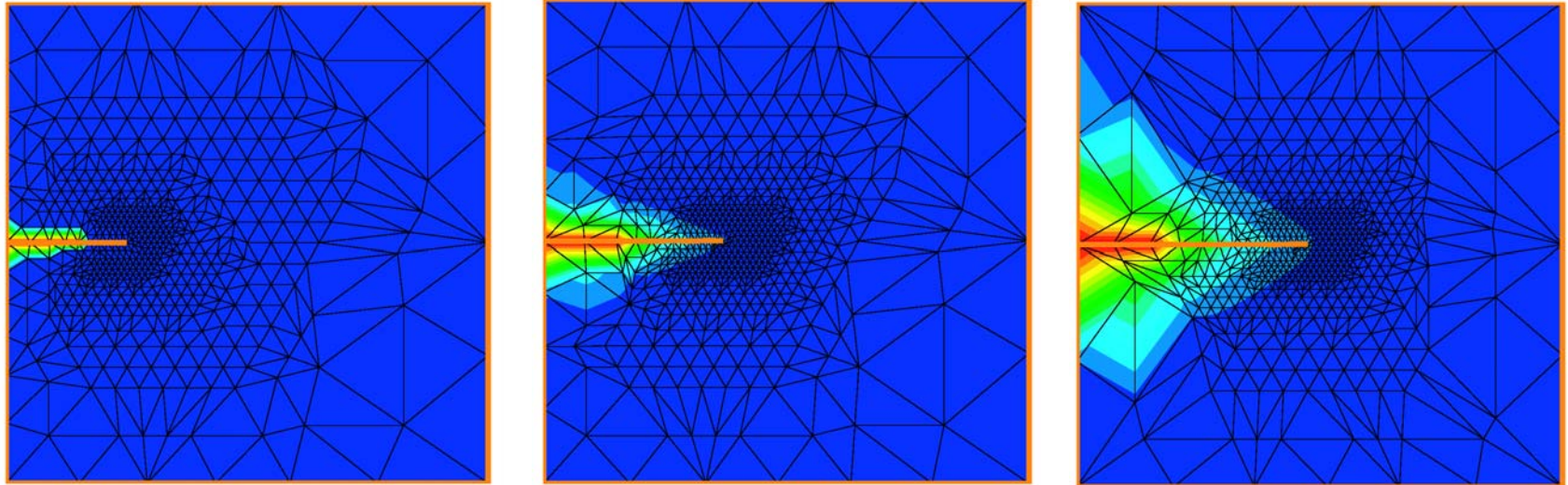


Center crack panel geometry.

- Material: Steel
  - $E = 207 \text{ GPa}$
  - $\nu = 0.3$
  - $\sigma_0 = 325 \text{ MPa}$
  - $G_c = 30.0 \text{ kJ/m}^2$
  - $\sigma_c = 840 \text{ MPa}$
  - $D = 1.27\text{e-}8 \text{ m}^2/\text{s}$
- Load.
  - $P = 260 \text{ MPa (Constant)}$ .
- Initial crack size
  - $a = 10 \text{ mm}$
- Impurity (hydrogen)
  - $V = 7.116\text{e-}6 \text{ m}^3 / \text{mol}$
  - $\Delta V = 2\text{e-}6 \text{ m}^3 / \text{mol}$
  - $C_0 = 2.084\text{e}21 \text{ atoms/ m}^3$

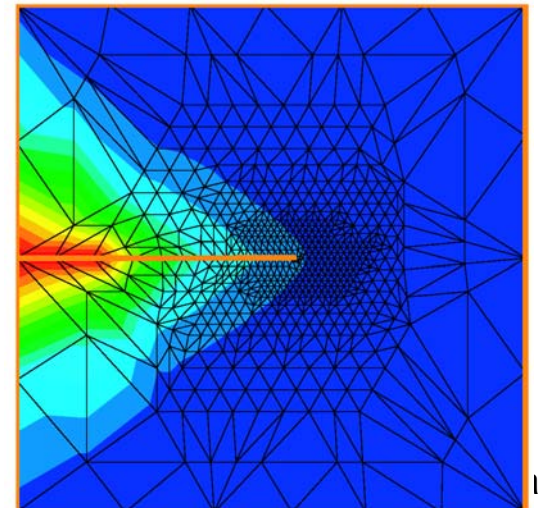


# SCC – Experimental validation



Contours of hydrogen concentration  
(Nguyen and Ortiz, 2000)

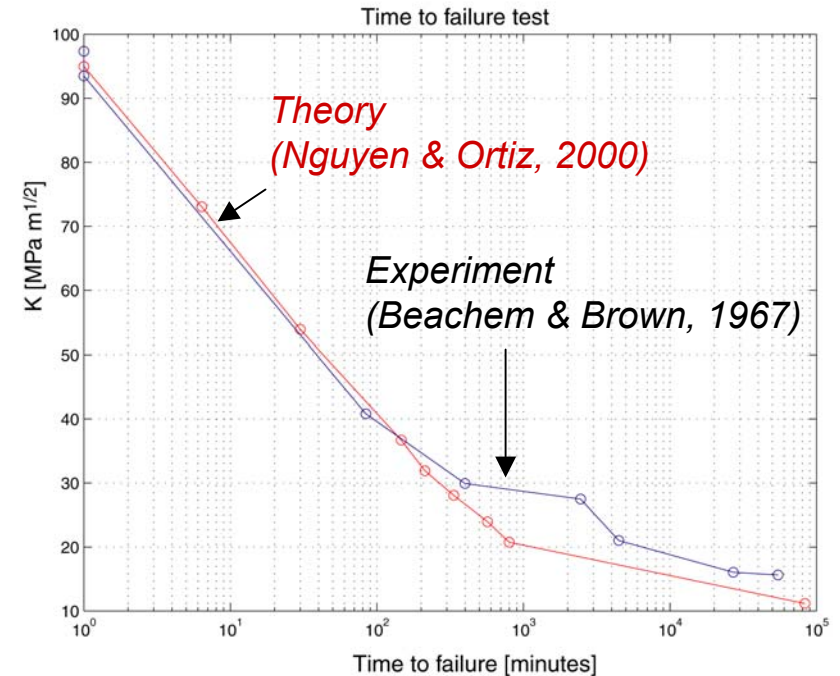
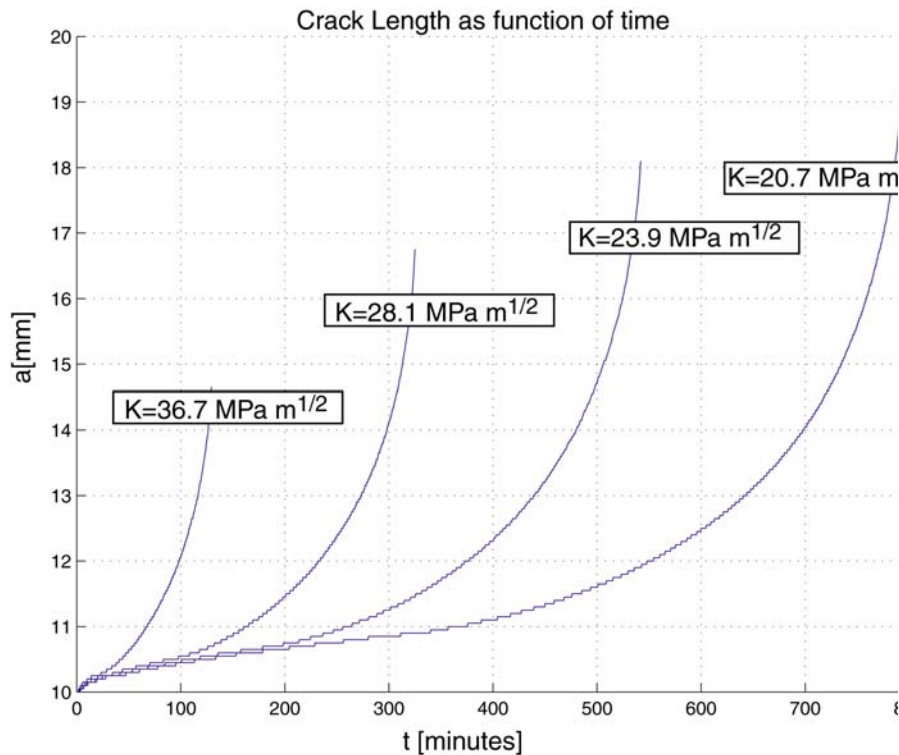
(Movie)



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# SCC - Experimental validation



(Nguyen and Ortiz, 2000)





# Multiscale Modeling of Fracture

- Engineering models of material behavior are fraught with **uncertainty**, empiricism
- The only way to remove uncertainty is to model the underlying physical mechanisms directly
- Often unit mechanisms occur over multiple scales simultaneously and bear a hierarchical relationship:  
**Multiscale modeling**
  - *Macro* → *micro*: *Driving forces*
  - *Micro* → *macro*: *Averaging*
- Lengthscale hierarchy stops at the atomistic level
- Application to fracture: Use atomic-scale binding energy relations calculated *ab initio*



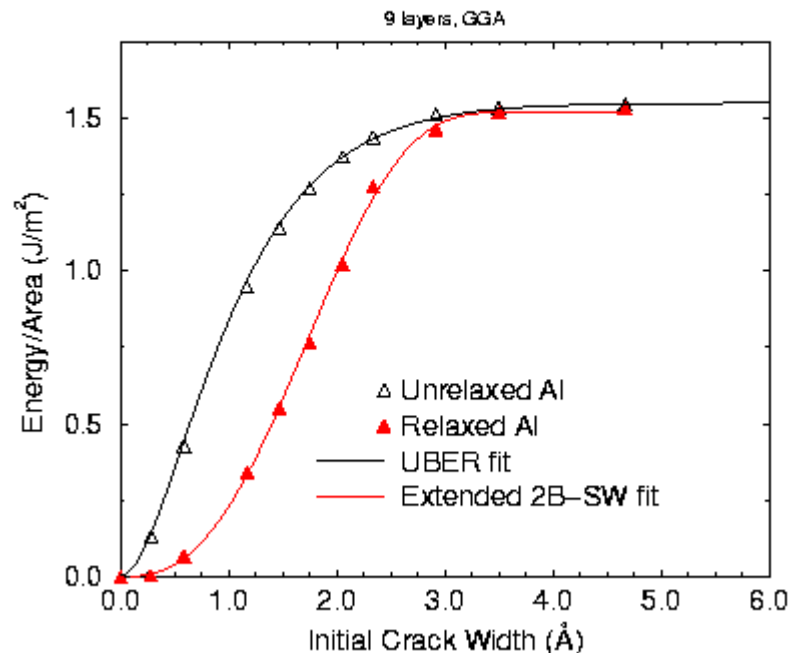


# Multiscale Modeling of Fracture

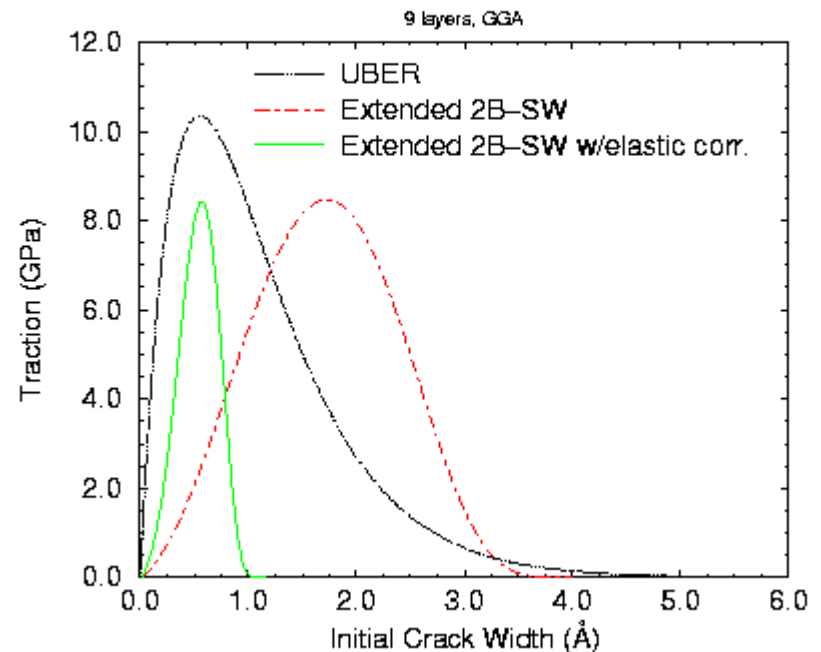
## Ab-initio studies of decohesion in Aluminum and Alumina

(EAA Jarvis, RL Hayes and EA Carter, *ChemPhysChem*, 2000)

Energy vs. Separation for (111) fcc Al



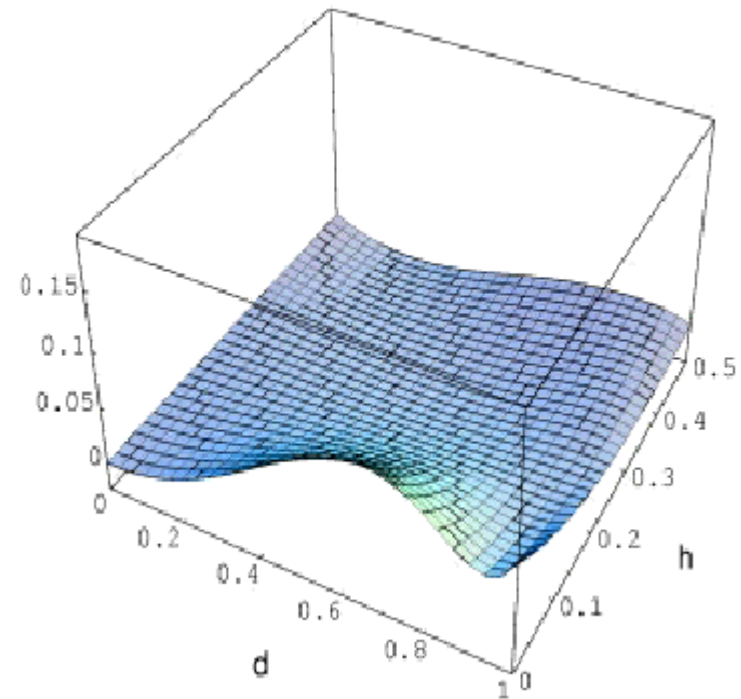
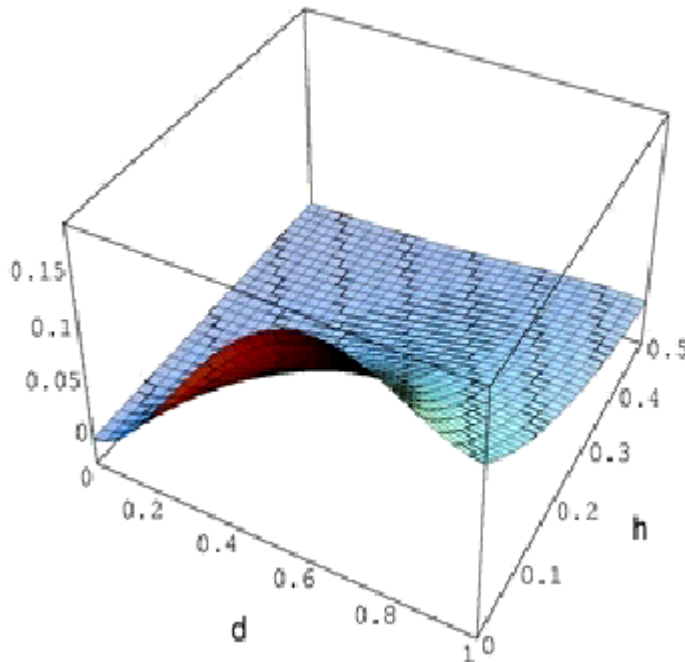
Traction vs. Separation for (111) fcc Al



# Multiscale Modeling of Fracture

## Ab-initio studies of Hydrogen embrittlement in Aluminum (Park and Kaxiras, 2000)

- Without Hydrogen:
- With Hydrogen:



Calculated stacking-fault surfaces  
as functions of interplanar sliding ( $d$ )  
and interplanar separation ( $h$ )



# Multiscale Modeling: The Chasm

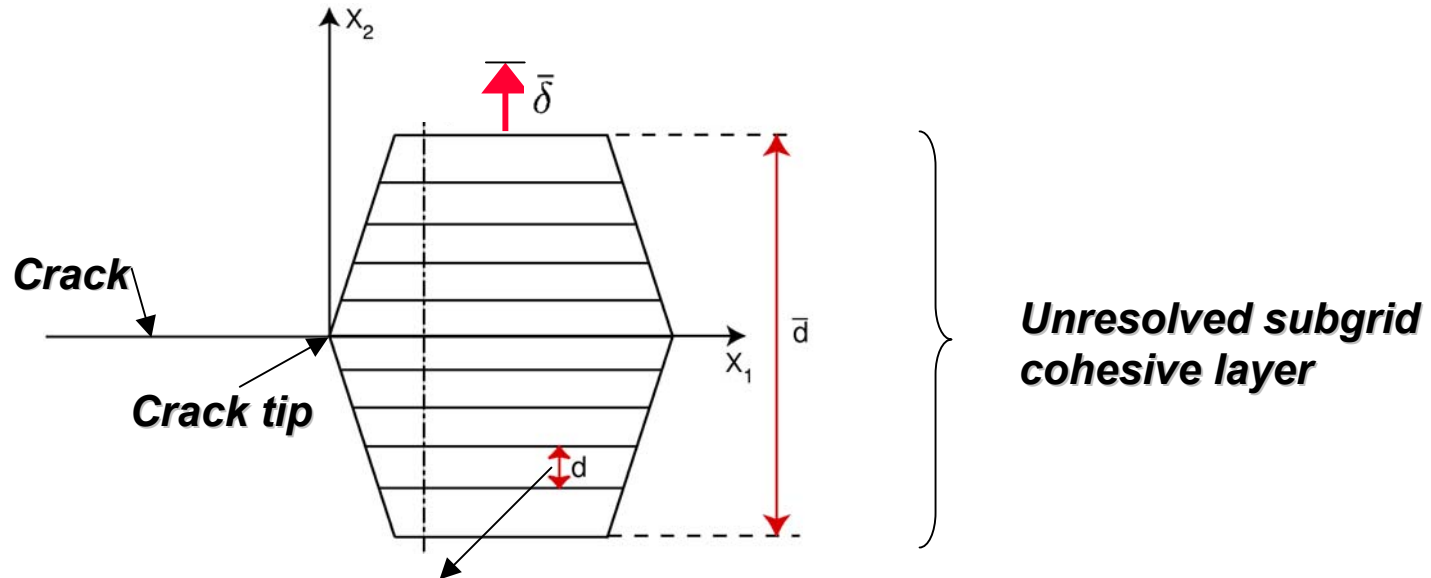
- Cohesive-zone size:  $R \sim E' G_c / \sigma_c^2$
- Typical cohesive-zone size from atomic-scale binding energy relations:

$$R \sim 1 \text{ nm}$$

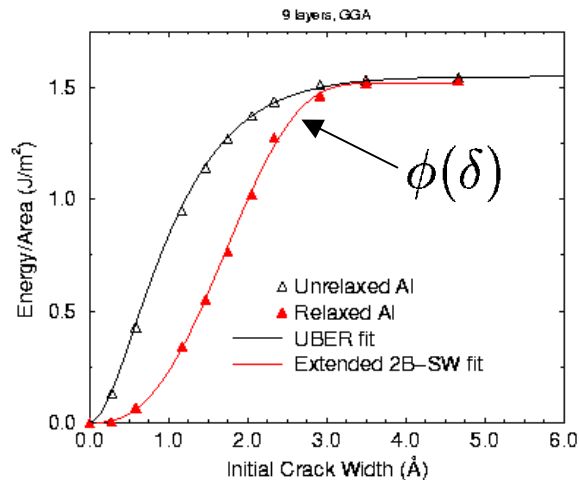
- This is beyond resolution afforded by macroscopic simulations: **Need to coarse-grain atomistic laws**
- Resolution gaps and techniques to address them:
  - *Turbulence: Subgrid models*
  - *Shock physics: Artificial viscosity models*
  - *Solids with microstructure: Theories of effective behavior*
    - *Weak convergence, energy relaxation*
    - *Statistical mechanics, renormalization group*



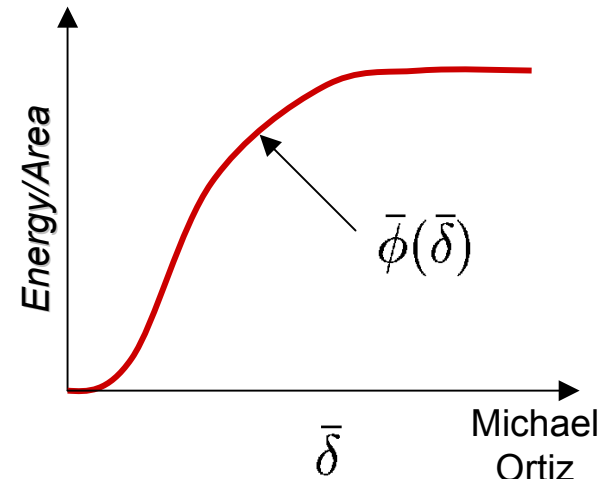
# Renormalization of *ab initio* BER



Energy vs. Separation for (111) fcc Al

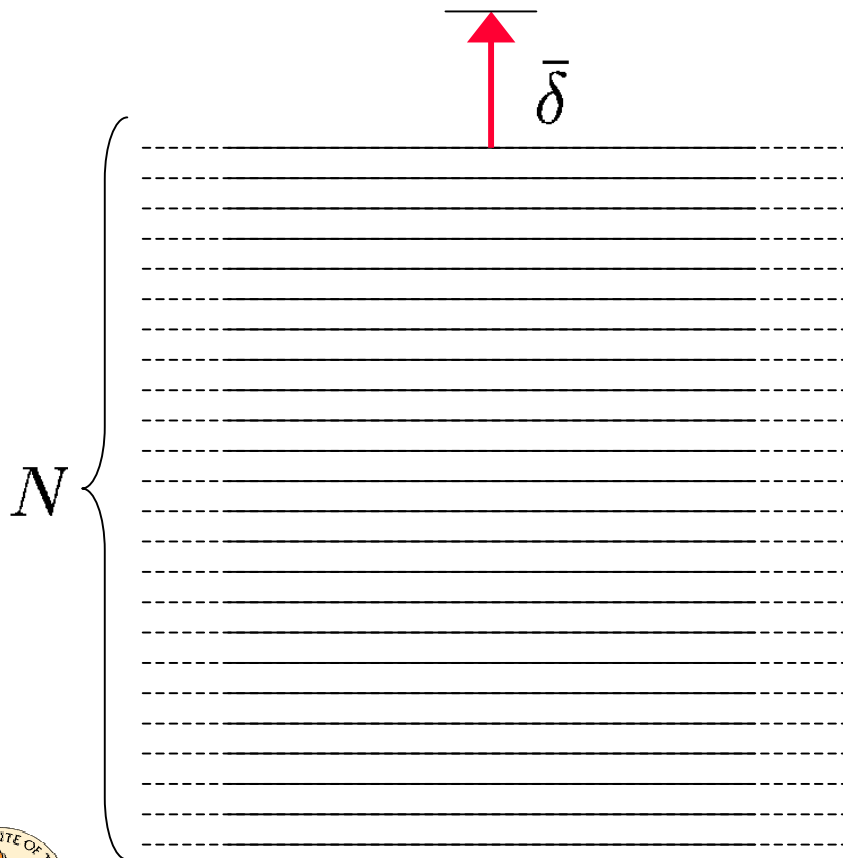


$$N = \bar{d}/d \gg 1$$



# Renormalization & energy minimization

- Operating principle: Energy minimization,



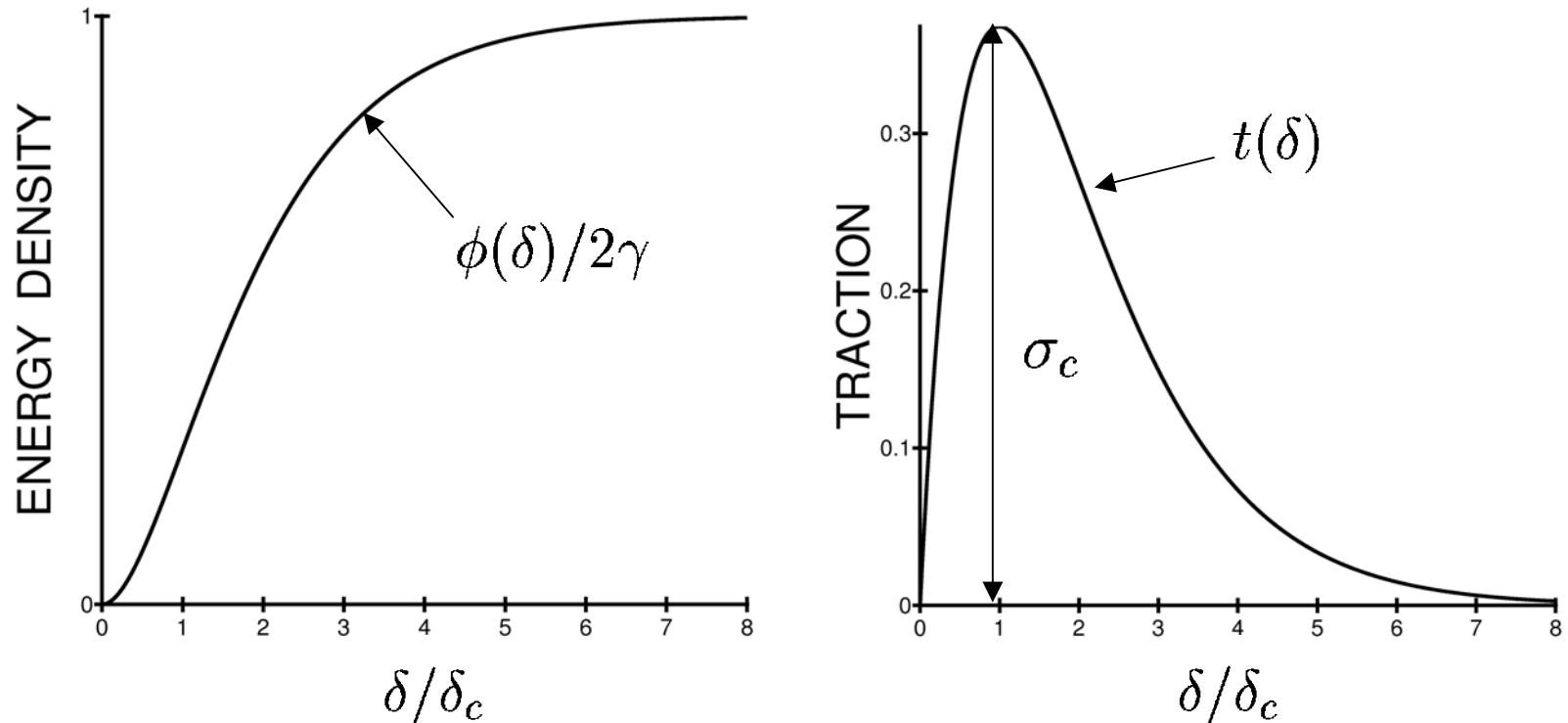
$$E^{\text{tot}} = \sum_{i=1}^N \phi(\delta_i)$$

$$\bar{\phi}(\bar{\delta}) = \inf_{\{\delta_1, \dots, \delta_N\}} \sum_{i=1}^N \phi(\delta_i)$$

$$\bar{\delta} = \sum_{i=1}^N \delta_i$$



# Renormalization - Assumptions



- Assumptions: Interatomic potential continuous, nondecreasing,

i)  $\phi(0) = 0$ ,

ii)  $\phi \rightarrow 2\gamma$  as  $\delta \rightarrow \infty$ ,

iii)  $\phi \sim (C/2)\delta^2 + o(\delta^2)$ ,  $C = (1/d)c_{ijkl}m_i m_j m_k m_l > 0$



# Renormalization – Asymptotic analysis

**Theorem:** Let  $\phi(\delta) : [0, \infty) \rightarrow [0, 2\gamma]$  be continuous, nondecreasing,  $\phi(0) = 0$ ,  $\phi \rightarrow 2\gamma$  as  $\delta \rightarrow \infty$ , and let

$$\lim_{\delta \rightarrow 0^+} \frac{\phi(\delta)}{\delta^2} = \frac{C}{2}, \quad C > 0$$

Then:

$$\bar{\phi}(\bar{\delta}) \sim \min\left\{\frac{\bar{C}}{2}\bar{\delta}^2, 2\gamma\right\} = \begin{cases} (\bar{C}/2)\bar{\delta}^2, & \text{if } \bar{\delta} < \bar{\delta}_c \\ 2\gamma, & \text{otherwise} \end{cases}$$

asymptotically as  $N \rightarrow \infty$ .

Effective parameters:

$$\bar{\delta}_c = 2\sqrt{\frac{\gamma N}{C}}, \quad \bar{\sigma}_c = 2\sqrt{\frac{C\gamma}{N}}, \quad \bar{C} = \frac{C}{N}$$





# Renormalization – Asymptotic analysis

- Equivalent reformulation of the problem:

$$\bar{\phi}(\bar{\delta}) = \inf_{\{(\delta_1, \delta_2), (N_1, N_2)\}} \{N_1 \phi(\delta_1) + N_2 \phi(\delta_2)\}$$

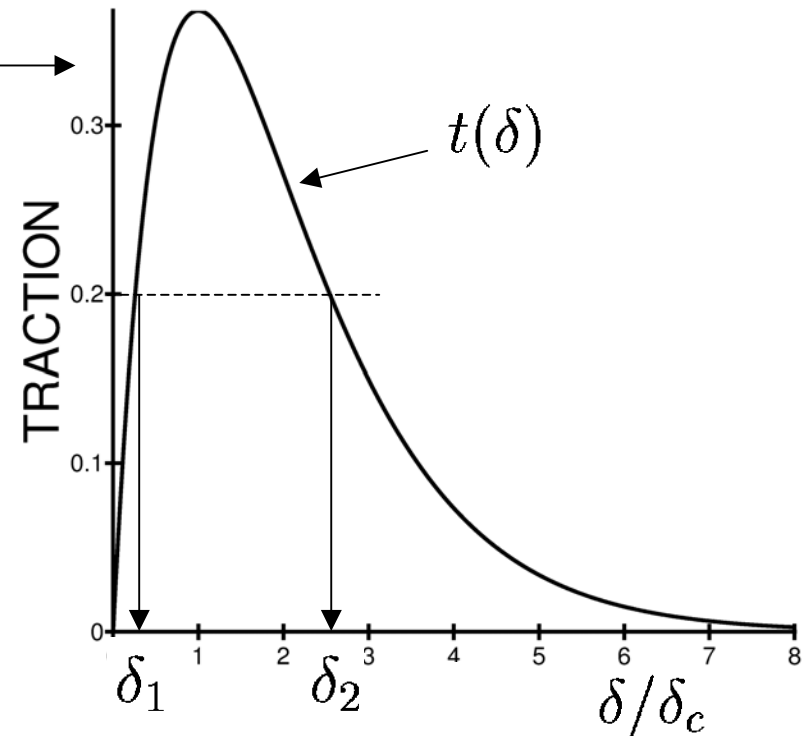
$$t(\delta_1) = t(\delta_2) \longrightarrow$$

$$\bar{\delta} = N_1 \delta_1 + N_2 \delta_2$$

$$N = N_1 + N_2$$

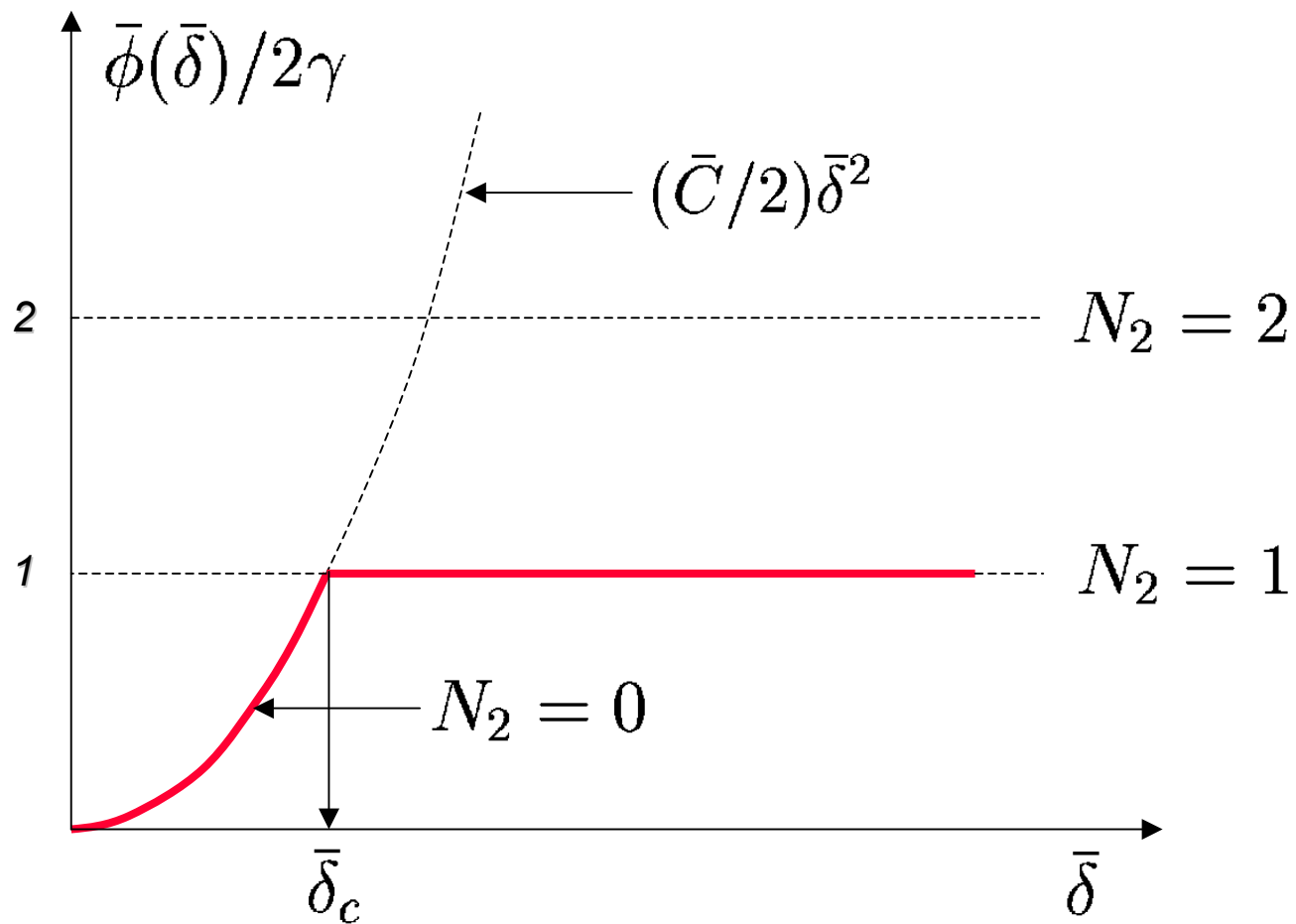
$$0 \leq \delta_1 < \delta_c$$

$$\delta_2 > \delta_c$$



# Renormalization – Asymptotic analysis

- Asymptotically, as  $N \rightarrow \infty$ ,



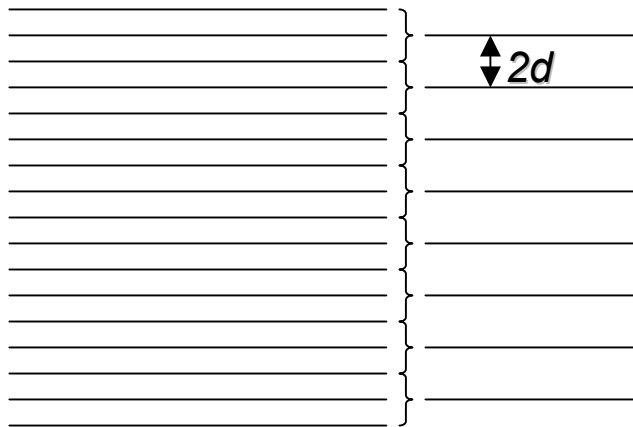
# Renormalization of *ab initio* BER

- Cohesive-layer potential attains a *universal asymptotic form* (parabolic + constant), independently of the form of the atomistic BER
- The renormalized peak stress scales as:  $1/\sqrt{N}$
- The renormalized critical opening displacement scales as:  $\sqrt{N}$
- Surface energy is preserved under renormalization
- The only information from the atomistic BER which 'survives' renormalization is:
  - *Elastic moduli*
  - *Surface energy*
- The renormalized cohesive zone size scales as:  $R \sim \bar{d}$



# The renormalization group

- Renormalization group for cohesive layer:



*Decimation*

$$\begin{aligned}\tilde{\phi}(\delta) &= \inf_{\{\delta_1, \delta_2\}} \{ \phi(\delta_1) + \phi(\delta_2) \} \\ \delta &= \delta_1 + \delta_2 \\ (R\phi)(\delta) &= \tilde{\phi}(\sqrt{2}\delta)\end{aligned}$$

- Scaling: Set  $(R\phi)(\delta) = \tilde{\phi}(\lambda\delta)$ . Require elastic moduli to be preserved by the RG transformation:

$$R\phi = \phi, \quad \forall \phi \text{ quadratic iff } \lambda = \sqrt{2}$$



# RG – Properties of the transformation

- $R$  preserves monotonicity:  $\delta_1 < \delta_2 \Rightarrow R\phi(\delta_1) < R\phi(\delta_2)$
- $R$  preserves fracture energy:  $R\phi(\infty) = \phi(\infty) = 2\gamma$
- $R$  preserves elastic moduli:  $R\phi''(0) = \phi''(0) = C$
- $R$  preserves ordering:  $\phi < \psi \Rightarrow R\phi < R\psi$
- The function:  $\bar{\phi}(\bar{\delta}) \sim \min\{\frac{\bar{C}}{2}\bar{\delta}^2, 2\gamma\}$  is a fixed point of  $R$



# RG – Convergence to fixed point

**Theorem:** Let  $\phi(\delta) : [0, \infty) \rightarrow [0, 2\gamma]$  be continuous, nondecreasing,  $\phi(0) = 0$ ,  $\phi \rightarrow 2\gamma$  as  $\delta \rightarrow \infty$ , and

$$\lim_{\delta \rightarrow 0^+} \frac{\phi(\delta)}{\delta^2} = \frac{C}{2}, \quad C > 0$$

Let:

$$\bar{\phi}(\bar{\delta}) \sim \min\left\{\frac{\bar{C}}{2}\bar{\delta}^2, 2\gamma\right\} = \begin{cases} (\bar{C}/2)\bar{\delta}^2, & \text{if } \bar{\delta} < \bar{\delta}_c \\ 2\gamma, & \text{otherwise} \end{cases}$$

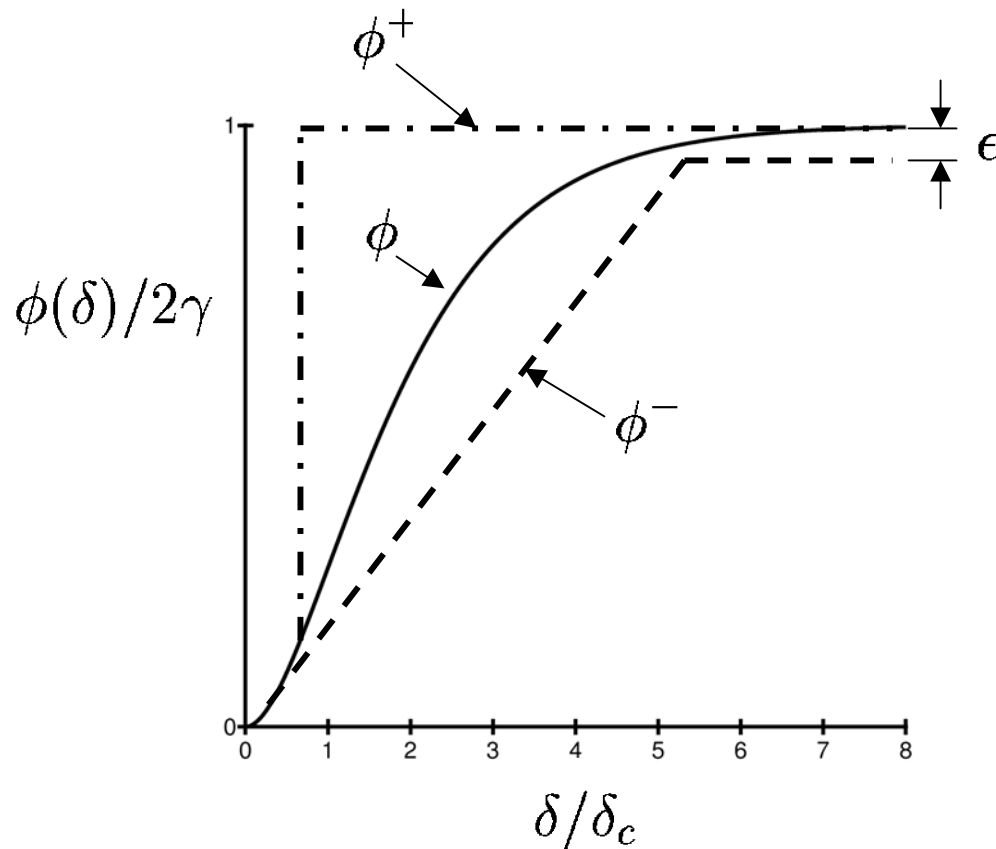
$\phi_0 = \phi$ , and

$$\phi_n = R\phi_{n-1} = R^n\phi_0, \quad n = 1, \dots, \infty$$

Then  $\phi_n \rightarrow \bar{\phi}$  uniformly in  $[0, \infty)$ .



# RG - Proof of convergence



$$R^n \phi^\pm \rightarrow \bar{\phi} \text{ uniformly in } [0, \infty) \Rightarrow R^n \phi \rightarrow \bar{\phi} \text{ uniformly in } [0, \infty)$$

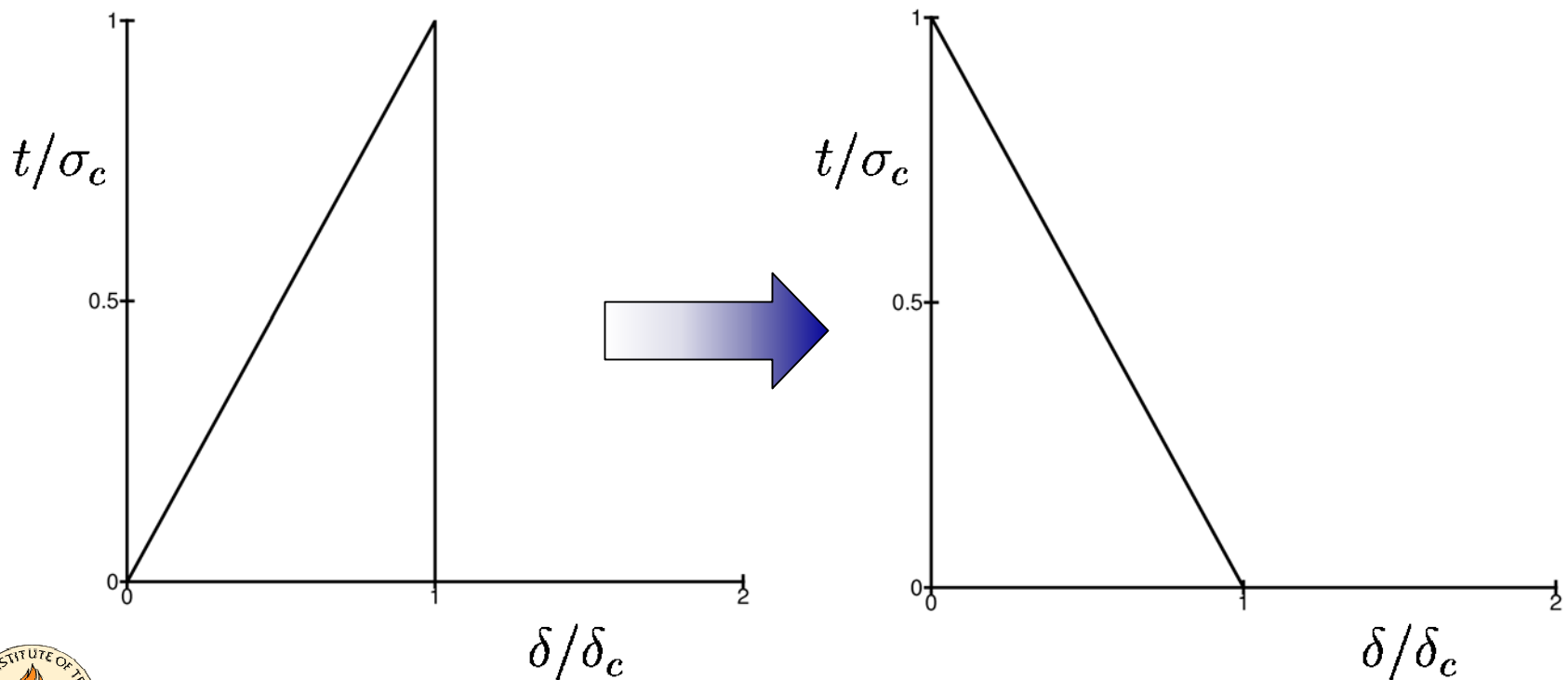
as  $n \rightarrow \infty$ ,  $\epsilon \rightarrow 0^+$





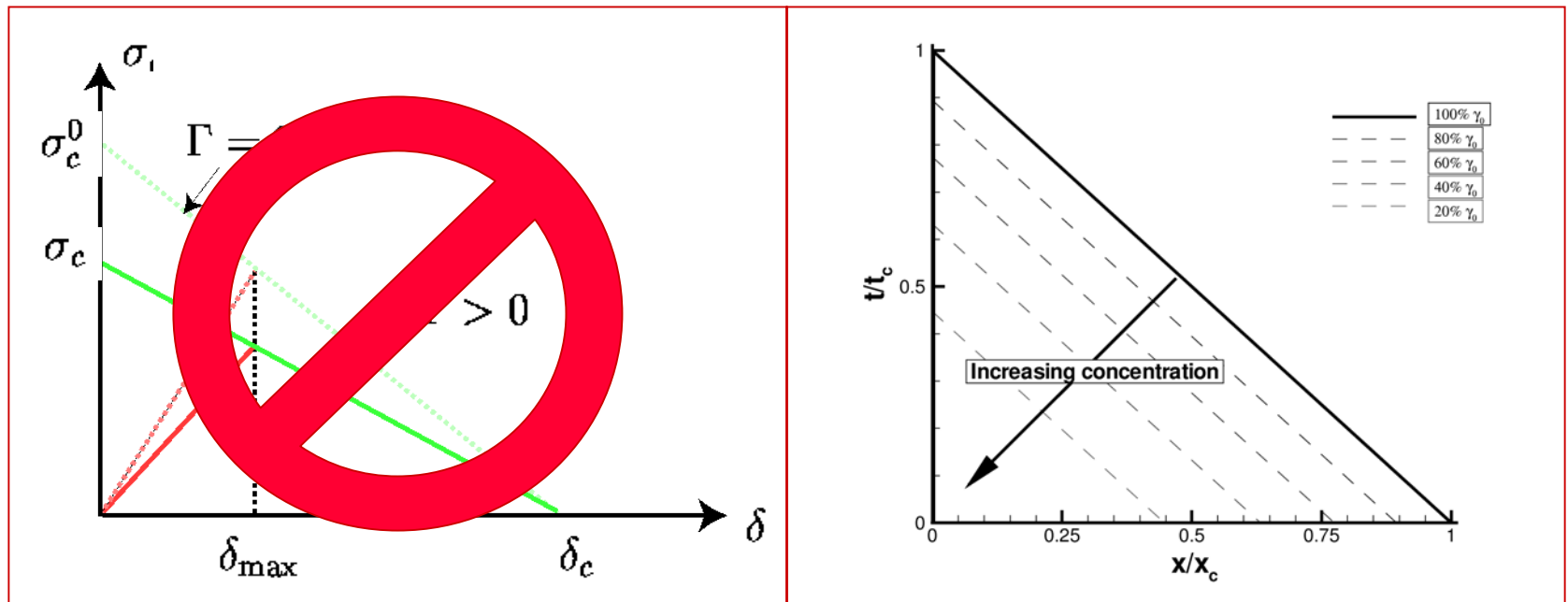
# Elastic correction - Cohesive law

- Need to perform correction to avoid double counting of elasticity of matrix (Rice, 1992; Beltz and Rice, 1992)

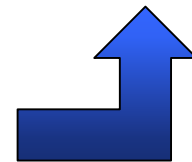


# Chemistry-dependent cohesive law

- Assumption: Elastic moduli independent of impurity concentration



$$\bar{\delta}_c = 2\sqrt{\frac{\gamma N}{C}}, \quad \bar{\sigma}_c = 2\sqrt{\frac{C\gamma}{N}}$$



# Concluding remarks

- Multiscale analysis can guide engineering modeling and help eliminate uncertainty
- The lengthscale gap between *ab initio* BERs and engineering models may effectively be bridged by (nonstandard) **renormalization**
- Renormalization leads asymptotically to a **universal** form of the macroscopic cohesive law
- Renormalized cohesive laws may be regarded as **subgrid models** which compensate for the lack of resolution of engineering models
- Connection/interaction to plasticity, microcracking, lattice defects, and other fracture mechanisms remains to be explored...



# Unsolved problems...



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